



Luminescence study of the LuBO₃ and LuPO₄ doped with RE³⁺

V.V. Mikhailin^{a,b}, D.A. Spassky^{a,*}, V.N. Kolobanov^b, A.A. Meotishvili^c, D.G. Permenov^c, B.I. Zadneprovski^c

^aSkobel'syn Institute of Nuclear Physics, M.V. Lomonosov Moscow State University, Vorob'evy Gory 1, bld. 2, 119991 Moscow, Russia

^bSynchrotron Radiation Laboratory, Physics Faculty, Moscow State University, 119991 Moscow, Russia

^cCentral Research and Development Institute of Chemistry and Mechanics, 115487 Moscow, Russia

ARTICLE INFO

Article history:

Received 14 August 2009

Received in revised form

22 November 2009

Accepted 13 December 2009

Keywords:

Lutetium phosphate

Lutetium borate

Luminescence

ABSTRACT

Lutetium borates and phosphates doped with RE³⁺ ions are perspective scintillators. In the paper, the results of the luminescence spectroscopy of LuPO₄ and LuBO₃ doped with Pr³⁺, Ce³⁺, Tb³⁺ and Eu³⁺ under synchrotron radiation excitation are presented. The processes of the energy transfer from the host lattice to the luminescence centers are considered. The creation of excitons at the edge of fundamental absorption region is shown. The bandgap values for lutetium borate and phosphate were estimated.

© 2009 Elsevier Ltd. All rights reserved.

1. Introduction

Borates and phosphates doped with rare-earth ions are considered as perspective materials for scintillating detectors, X-ray imaging and as plasma display panel phosphors (Balcerzyk et al., 2000; Rao and Devine, 2000). Incorporation of lutetium as a cation benefits in the increasing of material density and in high light yield in comparison to that with yttrium, gadolinium or lanthanum cations (Mansuy et al., 2004). The luminescence properties of LuPO₄ and LuBO₃ doped with RE³⁺ were widely investigated at the direct excitation of RE ions and at the high-energy excitation (X-rays). However, the luminescence properties at the excitation in the fundamental absorption region were poorly studied. Excitation spectra measured in this energy region contains information about the processes of energy transfer from the host lattice to the luminescence centers. This stage of relaxation is important for the efficient conversion of high-energy radiation into scintillations. The bandgap values (E_g) and information about excitons creation are also derived from the excitation spectra. Available data on the values of E_g are presented in Table 1. The results of the luminescence study of LuPO₄ and LuBO₃ doped with RE³⁺ ions (RE = Ce, Tb, Eu and Pr) are presented in the paper. Attention was focused on the processes of energy transfer from the host lattice to the luminescence centers as well as to the exciton creation on the edge of the fundamental absorption region.

2. Experimental details

LuPO₄ and LuBO₃ were synthesized by sol-gel method and characterized by the X-ray diffraction and scanning electron microscopy. X-ray phase analysis showed that the measured sample of LuBO₃ was crystallized in the vaterite structure. LuPO₄ crystallized in xenotime structure. Concentration of RE ions was 5 mol % for Tb, Eu and Pr and 0.5 mol% for Ce. Predominant size of the obtained crystalline particles varies in the range 250–1000 nm.

Spectroscopic study of the samples was carried out using synchrotron radiation in the energy region 3.7–22 eV at the SUPERLUMI station, DESY (Zimmerer, 2007). Synchrotron radiation is an ideal light source for the luminescence spectroscopy in VUV region due to its continuous spectrum, high intensity and temporal structure.

3. Results and discussion

Luminescence spectra of LuPO₄ doped with Tb, Eu and Pr as well as of LuBO₃ doped with Ce and Tb are presented in Fig. 1. Radiative transitions in RE ions determine the structure of luminescence spectra. Doublet in LuBO₃:Ce with maxima at 390 and 420 nm is due to the $5d-^2F_{5/2,7/2}$ transitions in Ce³⁺. In LuPO₄:Tb and LuBO₃:Tb, four intensive groups of peaks at 480–500, 535–560, 580–595 and 615–635 nm are due to the $^5D_3-^7F_J$ ($J = 3-6$) transitions in Tb³⁺. Three prominent groups of peaks were observed in LuPO₄:Eu at 585–600, 610–625 and 690–710 nm. These peaks are caused by $^5D_0-^7F_{1,2,4}$ transitions in Eu³⁺. Peaks in LuPO₄:Pr are related to Pr³⁺ $4f5d-4f^2$ [3H, 3F] transitions in the range

* Corresponding author. Tel.: +7 495 9393169; fax: +7 495 9392991.

E-mail address: deris2002@mail.ru (D.A. Spassky).

Table 1
Physical properties of LuPO₄ and LuBO₃.

	LuPO ₄	LuBO ₃
Structural type	Xenotime	Vaterite (V)/Calcite (C)
ρ , g/cm ³	6.53	7.42 (V) 6.87 (C)
Bandgap value, eV	8.7 ^a 8.6 ^b 8.85 ^c	7.0 ^b (V) 6.4 ^b (C)

^a Data on the bandgap values were taken from Lempicki and Wojtowicz, 1994.

^b Data on the bandgap values were taken from Balcerzyk et al., 2000.

^c Data on the bandgap values were taken from van Pieterse et al., 2002.

220–300 nm and ³P_{0,1}–³H_{4,6}, ³F_{2,4} transitions in the range 350–750 nm. For the latter sample, the additional peak at 313 nm was also observed that is the evidence of the presence of Gd³⁺ traces in this sample.

The excitation energy (10.8 eV) definitely exceeds the bandgap values of the crystals providing an opportunity for energy transfer to the intrinsic or defect-related luminescence centers. Actually for the LuBO₃ and LuPO₄ intrinsic luminescence is observed as broad bands with maxima at 360 nm and 280–300 nm, respectively (Duan et al., 2005; Wisniewski et al., 2002). These bands are not observed in Fig. 1 implying efficient energy transfer to the dopant luminescence centers even via interband excitation of the crystals.

The luminescence intensity was compared by integrating of the luminescence signal overall the spectral region of the measurements (see the inset to Fig. 1). The highest luminescence intensity was detected for the LuBO₃:Tb. For phosphates Tb-doped sample also demonstrates the highest luminescence intensity.

The excitation spectra of RE-luminescence for LuPO₄ crystals are presented in Fig. 2. Sharp lines with low intensity in the region 3–5 eV are ascribed to the parity-forbidden f-f transitions in RE ion (Mansuy et al., 2006). With further increase of excitation energy up to 8.5 eV, intensive groups of peaks ascribed to f-d parity-allowed transitions in Tb³⁺ and Pr³⁺ are observed (van Pieterse et al., 2002). For Eu³⁺, the charge-transfer state lies at lower energies than 4f⁵5d state thus the broad peak at 5.8 eV corresponds to the luminescence excitation via charge-transfer state (Dorenbos, 2005). Parity-allowed f-d transitions in LuPO₄:Eu³⁺ are manifested

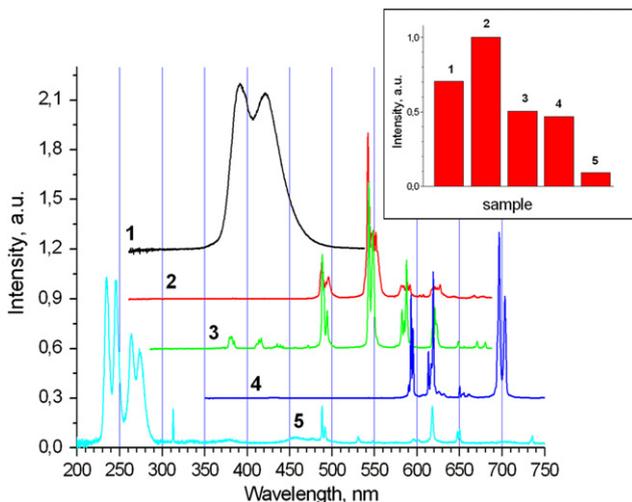


Fig. 1. Luminescence spectra of LuBO₃:Ce (1); LuBO₃:Tb (2); LuPO₄:Tb (3); LuPO₄:Eu (4) and LuPO₄:Pr (5) at $E_{ex} = 10.8$ eV, $T = 300$ K. In the inset: integrated intensity of the luminescence spectra.

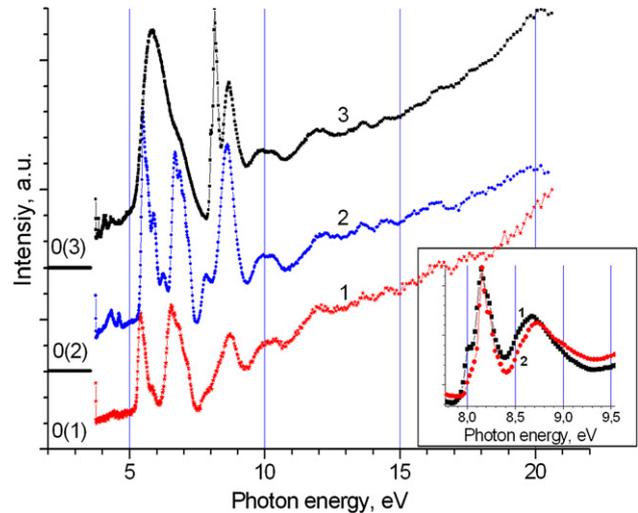


Fig. 2. Luminescence excitation spectra of LuPO₄:Eu ($\lambda_{em} = 595$ nm, curve (1), LuPO₄:Tb ($\lambda_{em} = 545$ nm, curve 2) and LuPO₄:Pr ($\lambda_{em} = 495$ nm, curve 3), $T = 300$ K. In the inset – excitation of LuPO₄:Eu in the exciton region at $T = 300$ K (curve 1) and $T = 10$ K (curve 2).

at higher energies as a complex peak at 8.15 eV (van Pieterse et al., 2002).

From the $E > 8.5$ eV, the excitation spectra demonstrate similar features that are independent to the dopant. The peaks at 8.7, 9.8, 12.0 eV, and the dips at 9.3, 10.7 eV perfectly coincide for all samples of LuPO₄. Similar behavior indicates that at energies $E > 8.5$ eV the luminescence is excited via the fundamental absorption region of the crystal. Several factors affect on the structure of spectra in this region. It is the peculiarities of electron transitions in the host lattice that are determined by the origin and density of electronic states of the valence and conduction bands of the crystal. Furthermore the observed spectra may be modulated with the losses on reflectivity and near-surface losses. Also the creation of exciton is possible at the edge of fundamental absorption region.

Time-resolved spectroscopy allows to separate the slow processes of luminescence excitation connected with sequential capture of separated charge carriers on the RE ion and the fast processes connected with direct excitation of the luminescence center or via exciton creation. In Fig. 3, the luminescence excitation spectra of LuPO₄:Pr measured in “fast” (3–16 ns after the excitation pulse of synchrotron radiation) and “slow” (120–155 ns) time windows are presented. The time windows values were taken assuming the decay time of Pr luminescence at the intracenter f-d excitation of Pr ion (the decay is fitted with two exponentials with $\tau_1 = 3.4$ ns and $\tau_2 = 12$ ns). Obviously, the peaks that were attributed to the parity-allowed f-d transitions within Pr ion are observed in the spectrum measured with “fast” time windows. However, the peak at 8.7 eV preliminary attributed to the excitation of the host lattice is also observed in the “fast” time windows. We suppose that the peak is of exciton origin that provides fast transfer of correlated electron-hole pair to the luminescence centers. Another indication on the exciton origin of the peak at 8.7 eV is its temperature dependence studied for LuPO₄:Eu. Its maximum shifts to the high-energies of 0.15 eV with the temperature decrease from 300 to 10 K (see the inset Fig. 2) obeying Urbach rule. At the same time the shift of the peak 8.15 eV that is due to f-d excitation in Eu³⁺ is negligible.

Noticeable rise of luminescence in excitation spectrum measured in the “slow” time windows starts at energies exceeding the exciton peak – 8.8 eV. The rise of slow component is connected

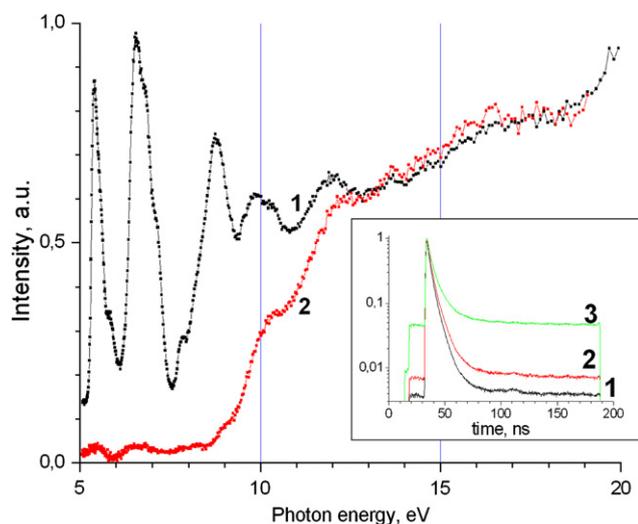


Fig. 3. Time-resolved luminescence excitation spectra of LuPO₄:Pr ($\lambda_{em} = 265$ nm) measured in the “fast” (3–16 ns, curve 1) and “slow” (120–155 ns, curve 2) time windows. The intensity of the spectra is normalized. In the inset – luminescence decay curves at $E_{ex} = 5.4$ –6.7 eV (curve 1), 8.85 eV (curve 2) and 10.8 eV (curve 3).

with the creation of separated electrons and holes in the conduction and valence bands of LuPO₄. The relative contribution of separated charge carriers into the luminescence excitation gradually increases up to 12 eV demonstrating competition of this mechanism with the excitation via exciton creation. The rise at 8.8 eV can be used for the estimation of the bandgap value of LuPO₄. This value agrees with that available from the literature (see Table 1, reference c).

Luminescence excitation spectra of LuBO₃:RE (RE = Tb, Ce) are presented in Fig. 4. The structure in the energy region up to 7.2 eV is connected with excitation via f-d parity-allowed transitions. From the peak at 7.85 eV the main features in structure in the excitation spectra of LuBO₃ doped with Tb and Ce ions are the same. Analogously to phosphates the exciton is detected at the edge of the fundamental absorption region using time-resolved spectroscopy.

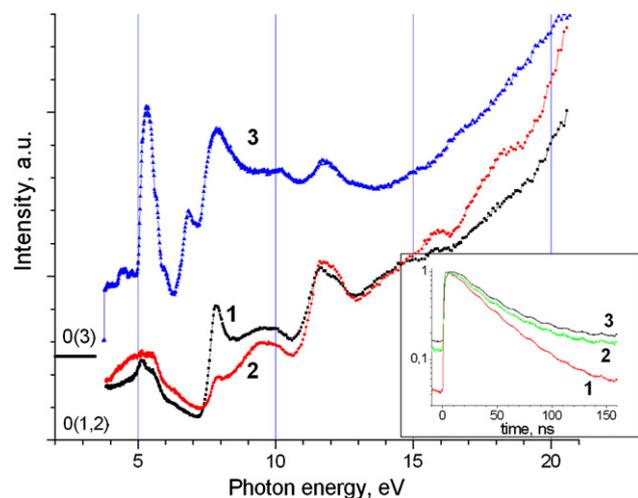


Fig. 4. Time-resolved luminescence excitation spectra of LuBO₃:Ce ($\lambda_{em} = 425$ nm) measured in the “fast” (3–20 ns, curve 1) and “slow” (90–155 ns, curve 2) time windows as well as luminescence excitation spectrum of LuBO₃:Tb ($\lambda_{em} = 545$ nm, curve 3). In the inset – luminescence decay curves at the $E_{ex} = 7.75$ eV (curve 1), 5.16 eV (curve 2) and 10.8 eV (curve 3).

The separation between the slow and fast processes is worse for LuBO₃:Ce than for LuPO₄:Pr due to slower decay time of Ce³⁺ luminescence. At excitation in exciton peak (7.75 eV), decay time was 55 ns. Slow decay components appear at the lower (5.16 eV) and higher (10.8 eV) excitation energies. In the latter case, luminescence rise-on is also observed. Presence of the slow decay component at excitation in the region of f-d transition within Ce³⁺ is not clear. Overlapping of cerium states with some defect states may be the reason of decay slowing, however, this supposition requires further improvement.

Relatively slow decay time of LuBO₃ complicates the determination of the separated charge carriers creation edge in LuBO₃. However, the bandgap value of LuBO₃ estimated from the presented results ($E > 7.75$ eV) significantly exceeds the values available from the literature (see Table 1).

From the luminescence excitation spectra follows that the energy transfer from the host material to RE³⁺ ions is efficient both in lutetium borates and phosphates. The intensity of the luminescence at the interband excitation stays approximately at the same level for the borates from the exciton peak up to the edge of photon multiplication process 17 eV (Zhang et al., 1999). For LuBO₃:Tb, the intensity is only slightly lower to that at the direct excitation of Tb³⁺ ion. In the case of phosphates, the decrease of the luminescence intensity was observed at the energies exceeding the exciton peak. It allows to suppose the existence of other (non-radiative) relaxation channel. In our measurements, it appeared in lower values of intensity for Tb-doped phosphate in comparison to borate (inset to Fig. 1) at the interband excitation. Moreover, it may be the one of the reasons for the lower value of light output from the lutetium phosphates in comparison with borates (Balcerzyk et al., 2000; Mansuy et al., 2004).

4. Conclusions

Luminescence properties of LuPO₄ and LuBO₃ doped with RE³⁺ (RE = Ce, Tb, Eu, Pr) were studied. Luminescence is related to the RE³⁺ emission without noticeable contribution of the intrinsic luminescence. The highest luminescence intensity was detected for LuBO₃:Tb at the interband excitation. Exciton creation in LuPO₄ and LuBO₃ at the edge of fundamental absorption region is demonstrated. Bandgap values were estimated using time-resolved spectroscopy. Efficient energy transfer from the host lattice to luminescence centers is one of the reasons for the superior scintillating properties of Lu-based borates and phosphates.

Acknowledgments

The support of grants RFBR 08-03-13511 and DFG 436 RUS 113/437 is gratefully acknowledged. We are grateful to Prof. G. Zimmerer for providing the opportunity to perform measurements at the SUPERLUMI station and to Dr. A. Kotlov for his help during measurements.

References

- Balcerzyk, M., Gontarz, Z., Moszynski, M., Kapusta, M., 2000. Future hosts for fast and high light output cerium-doped scintillator. *J. Lumin.* 87–89, 963–966.
- Dorenbos, P., 2005. The Eu³⁺ charge transfer energy and the relation with the band gap of compounds. *J. Lumin.* 111, 89–104.
- Duan, C., Yuan, J., Zhao, J., 2005. Luminescence properties of efficient X-ray phosphors of YBa₃B₉O₁₈, LuBa₃(BO₃)₃, a-YBa₃(BO₃)₃ and LuBO₃. *J. Sol. St. Chem.* 178, 3698–3702.
- Lempicki, A., Wojtowicz, A.J., 1994. Fundamental limitations of scintillators. *J. Lumin.* 60/61, 942–947.
- Mansuy, C., Nedelec, J.M., Mahiou, R., 2004. Molecular design of inorganic scintillators: from alkoxides to scintillating materials. *J. Mater. Chem.* 14, 3274–3280.

- Mansuy, C., Nedelec, J.M., Dujardin, C., Mahiou, R., 2006. Scintillation of sol-gel derived lutetium orthophosphate doped with rare earth ions. *J. Sol-Gel Sc. Tech.* 38, 97–105.
- van Pieterse, L., Reid, M.F., Wegh, R.T., Soverna, S., Meijerink, A., 2002. $4f^n-4f^{n-15}d$ transitions of the light lanthanides: experiment and theory. *Phys. Rev. B.* 65 (045113), 045114.
- Rao, R.P., Devine, D.J., 2000. RE-activated lanthanide phosphate phosphors for PDP applications. *J. Lumin.* 87–89, 1260–1263.
- Wisniewski, D., Tavernier, S., Wojtowicz, A.J., Wisniewska, M., Bruyndonckx, P., Dorenbos, P., van Loef, E., van Eijk, C.W.E., Boatner, L.A., 2002. LuPO₄:Nd and YPO₄:Nd—new promising VUV scintillation materials. *Nucl. Instr. Meth. A* 486, 239–243.
- Zhang, L., Pedrini, C., Madej, C., Dujardin, C., Gcon, J.C., Moine, B., Kamenskikh, I., Belsky, A., Shaw, D.A., MacDonald, M.A., Mesnard, P., Fouassier, C., Van't Spijker, J.C., Van Eijk, C.W.E., 1999. Fast fluorescence and scintillation properties of cerium and praseodymium doped lutetium orthoborates. *Rad. Eff. Def. Sol.* 150, 47–52.
- Zimmerer, G., 2007. SUPERLUMI: a unique setup for luminescence spectroscopy with synchrotron radiation. *Radiat. Meas.* 42, 859–864.